

RESEARCH

Open Access



Naturally-meaningful and efficient descriptors: machine learning of material properties based on robust one-shot ab initio descriptors

Sherif Abdulkader Tawfik^{1,2*} and Salvy P. Russo^{1,3*}

Abstract

Establishing a data-driven pipeline for the discovery of novel materials requires the engineering of material features that can be feasibly calculated and can be applied to predict a material's target properties. Here we propose a new class of descriptors for describing crystal structures, which we term Robust One-Shot Ab initio (ROSA) descriptors. ROSA is computationally cheap and is shown to accurately predict a range of material properties. These simple and intuitive class of descriptors are generated from the energetics of a material at a low level of theory using an incomplete ab initio calculation. We demonstrate how the incorporation of ROSA descriptors in ML-based property prediction leads to accurate predictions over a wide range of crystals, amorphized crystals, metal-organic frameworks and molecules. We believe that the low computational cost and ease of use of these descriptors will significantly improve ML-based predictions.

Introduction

A major objective in material science is to generate machine learning (ML) models that can accurately, and rapidly, predict a property for a given material by using information derived from the material's structure only [1, 2]. Predicting material properties such as the energy bandgap would then only take a few seconds or a fraction of a second using an ML model, instead of consuming several hours, or even days on a supercomputer to perform a first principles calculation, such as density functional theory (DFT). With the availability of massive materials datasets such as MaterialsProject [3] and AFLOW [4] which host more than 3.5 million materials, it is becoming increasingly possible to screen materials

for their properties [2]. To achieve such an objective, one must find features that can map a material structure against the highly nonlinear material properties. The vector of feature descriptors (the individual quantities that constitute the feature)¹ must be unique to each material and feasible to calculate. An ML model can subsequently be trained to translate descriptors into properties i.e. perform the mapping of structure against property. No matter how sophisticated or "deep" the ML models are, they will fail as long as the descriptors are poorly chosen.

¹ The terms "features" and "descriptors" are used interchangeably in the literature. Here we refer to a "feature" as a group of "descriptors." Note that other terms, including "attributes" and "fingerprints", are also frequently used in the literature, and they have the same meaning as "descriptors".

*Correspondence: sherif.tawfik@gmail.com; salvy.russo@rmit.edu.au

¹ ARC Centre of Excellence in Exciton Science, School of Science, RMIT University, Melbourne, VIC 3001, Australia
Full list of author information is available at the end of the article



The quality of descriptors is usually appraised by the ability of the descriptors to train predictive ML models. However, we emphasize the importance of three other key elements for judging the quality of descriptors, that are as important as their predictive power:

1. **Meaningfulness of features:** the term “meaningful features” appears frequently in the broad ML literature, such in in Ref. [5]. In the context of material science, the term loosely means that the features are related to a physical and/or chemical principle. An example is Ref. [6].
2. **Calculation efficiency:** the cost of computing the feature should be much less than that of calculating the target property.
3. **Number of descriptors within a feature:** the expression of a material structure into a relatively small number of features (i.e. a few hundreds) can ensure the simplicity of the ML model. Features that require the calculation of thousands of features entail costly storage requirements for datasets, higher processing requirements for the utilization of the trained ML models, and non-transparent, or “black-box” ML models [5].

We call these four criteria of ML features for materials the MENA criteria (Meaningful, Efficient, small Number of descriptors, Accurate). A number of ML features have recently been proposed in the literature for predicting the various DFT-calculated properties for materials, but they differ in how they satisfy the MENA criteria. We classify these features into the following four classes:

1. **Elemental features:** this is the simplest type of features, and the quickest to calculate. The descriptor values within these features are directly related to a property of the elements within the crystal structure or molecule, and therefore are physically and chemically meaningful. For example, for a crystal structure, a possible elemental feature is the mean atomic number and mean elemental melting point of the atoms within the crystal unit cell. However, these features are nonunique; two materials with equal composition, but different structural phases, will have the same elemental features. They are thus not accurate. It was also reported that that, in some cases, the most significant features for predicting a property seem to be counter-intuitive [6]. Using those features alone might work in limited cases, such as when using a small dataset (such as the ~300 materials dataset in Ref. [7]), but cannot be generalized for the broader set of materials. The reason these features work is related to the distribution of polymorphs in present-

day materials databases: in MaterialsProject, for example, the average number of polymorphs for each materials is ~1.4. If there were more polymorphs, elemental features might suffer from the fact that a material will have multiple property values (such as SiC, which has 27 polymorphs in MaterialsProject and their bandgaps range from 0 eV to 2.3 eV).

2. **Geometry-based features:** these include property-labelled materials fragments (PLMFs) [8], crystal graphs [9] and symmetry functions [10], among others. These features calculate translationally-invariant geometric, as well as elemental, quantities based purely on the material’s geometry. A simple, yet effective geometric descriptor is the symmetry group of the material’s lattice, which is ideally hot-coded into 230 separate zero-or-one columns. Many of the features in this class are mathematically very complex. An example of such descriptors is the symmetry functions. They are evaluated from the summation of exponential functions of the atomic distances. The features in this class are generally feasible to calculate.
3. **Electronic structure features:** they are calculated based on the electronic properties of the individual constituents of the material or molecule. That is, they are derived for separate atoms and/or bonds within the structure, but not for the entire structure. Examples include: the electronic structure attributes [11], molecular orbital attributes [12] and smooth overlap of atomic positions (SOAP) [13].
4. **Ab initio-based features:** examples are the molecular orbital energies [14] and the OrbNet Denali descriptors [15]. These descriptors are calculated by performing a full ab initio calculation at a low level of theory, and then the output of this calculation is used as descriptors for predicting the system’s properties at a higher level of theory; that is, ML here is “correcting” the outcome of low-level theory. A related ML procedure is the Δ -ML [16, 17], in which an ML model is trained to predict the difference in the value of a property, such as the HOMO–LUMO gap, between the value obtained using a high level of theory, and that using a low level of theory. Using ML to correct the results of DFT calculations has been known for a while [18]. The descriptors in these features are highly meaningful, since that they correspond directly to physically-computed quantities. In fact, the meaning of the descriptor values overlap with that of the target properties, such as the case of using molecular orbital energies (HOMO and LUMO values) to train a model to predict the HOMO–LUMO gap. However, this class of features is the most expensive to calculate; a DFT calculation

scales as N^3 , where N = number of atoms in the unit cell [19]. Thus, the calculation of the descriptors in these features typically require the utilization of high-performance computing facilities and long computing hours.

In this work we propose a new feature that can achieve the MENA criteria. Inspired by the ab initio-based features class, our *robust one-shot* ab initio (ROSA) descriptors of a given material are DFT-based descriptors. However, unlike the current member features of this class, ROSA descriptors are not calculated self-consistently; in fact, they are calculated by performing only one step in the self-consistent field (SCF) iteration. The ROSA descriptor values include the eigenvalues and total energy components that result from this computational step. ROSA descriptors are more computationally efficient than other ab initio-based features, equally meaningful, can be expressed with a small number of descriptors (109 descriptors, as will be explained below) and are highly accurate, as will be demonstrated in the following sections. We augment the ROSA descriptors with other material features, including atom-based and geometry features, and compare the predictive power of the different feature classes. We demonstrate the accuracy of predicting a range of material properties using the ROSA descriptors. These descriptors are also shown to be predictive for properties that are not directly related to a material's energy, such as the material's vibration properties. By using the energy bandgap, calculated using the popular Perdew-Burke-Ernzerhof (PBE) functional [20], as an additional input quantity, we also demonstrate the accuracy of predicting high-level energy properties of materials, namely: the HSE bandgap [21], GW bandgap [22] and the exciton binding energy calculated by solving the Bethe-Salpeter equation (BSE) [23]. We display a schematic diagram of the ROSA descriptors in Fig. 1a.

The manuscript is organized as follows: Section “**Features**” introduces the ROSA feature as well as the other features that will be used in this work, Section “**Results and discussion**” presents the results of training machine learning models on a range of material properties: Section “**Energetics, mechanical and vibrational properties of bulk systems**” is for energetic, mechanical and vibrational properties of bulk material properties, Section “**High-level energetics of two-dimensional materials**” is for the higher-level energetics of two-dimensional materials, and Section “**Prediction of properties of molecular systems**” is for the prediction of properties molecular systems. Section “**Conclusion**” is the conclusion of the work.

We examine the calculation CPU times and the calculated ROSA eigenvalues, based on single-core calculations performed using Intel Xeon Scalable processor

cores. For a sample of ~230 materials from the MaterialsProject with varying sizes, we compare the CPU time and the eigenvalues obtained by running a full SCF using VASP on a $4 \times 4 \times 4$ k-points mesh, the ROSA feature calculation using the linear combination of atomic orbitals (LCAO) (labelled ROSA LCAO) and the plane wave basis set (labelled ROSA PW), and the ROSA calculation using VASP version 5.4.4 [24]. All of these calculations have been performed on a single core, to enable direct comparison of CPU times. The results are displayed in Fig. 2a. In the figure, while the CPU time of ROSA PW and ROSA LCAO nearly coincide for a large number of materials, ROSA PW takes more CPU time than ROSA LCAO due to the increased complexity of PW calculations in larger unit cells. For this reason, the LCAO basis sets are used in the present ROSA calculations. Moreover, the full SCF calculation consumes much more time than both ROSA PW and ROSA LCAO by 1 to 4 orders of magnitude, which shows the significant time saving that is achieved with ROSA descriptors.

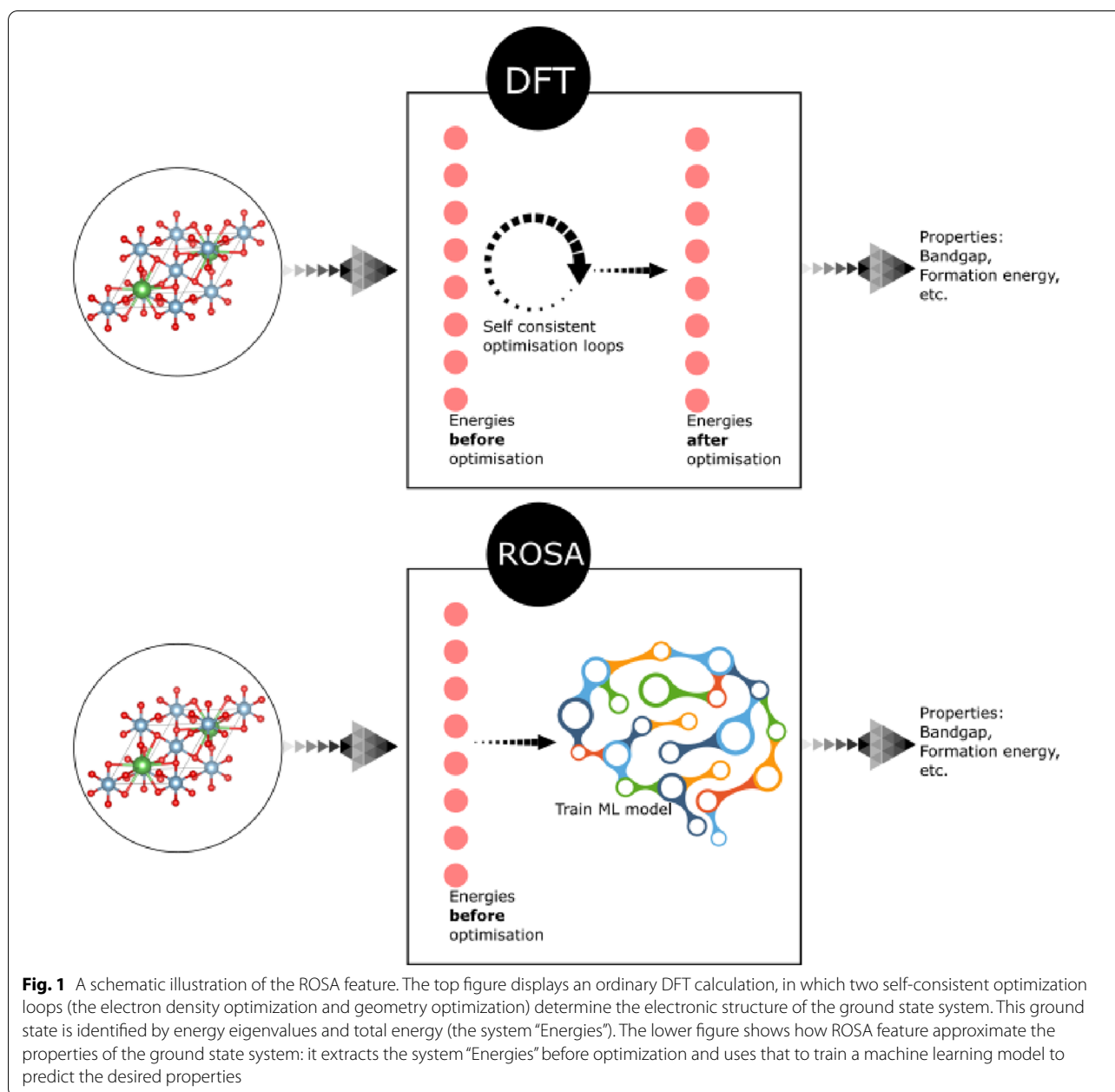
In Fig. 2b, we compare the eigenvalues obtained using the different methods by calculating the absolute value of the difference between the 100 eigenvalues obtained using ROSA LCAO, and those obtained using ROSA PW, VASP PW and VASP full SCF. The eigenvalues of ROSA PW are close to those of ROSA LCAO, while the eigenvalues obtained using VASP full SCF and VASP PW are quite different from those obtained using ROSA LCAO.

Features

In this work we have utilized three groups of features: the ROSA feature, the atom-based statistical properties, and a modification of the symmetry functions introduced in Ref [10].

ROSA feature

Although the calculation of the descriptors in the ROSA feature requires a DFT calculation, it is a rather “superficial” calculation, in which the applied DFT is set at the lowest possible level of accuracy, and only one non-self-consistent SCF step is executed; that is, not a full SCF iteration. The descriptors obtained from this calculation are the energy eigenvalues and total energies. We use the python-based GPAW code (version 21.1.0) [25], which can be easily installed on personal computers, for these calculations. The spin-restricted PBE exchange–correlation function is used, and calculations are performed at the Γ point. Note that setting the maximum iteration to 1 without raising an error from the present version of GPAW required that the source code be modified. This ROSA calculation produces a unique set of eigenvalues and 9 total energy values for each material. We centre each set of eigenvalues at the Fermi level and consider the

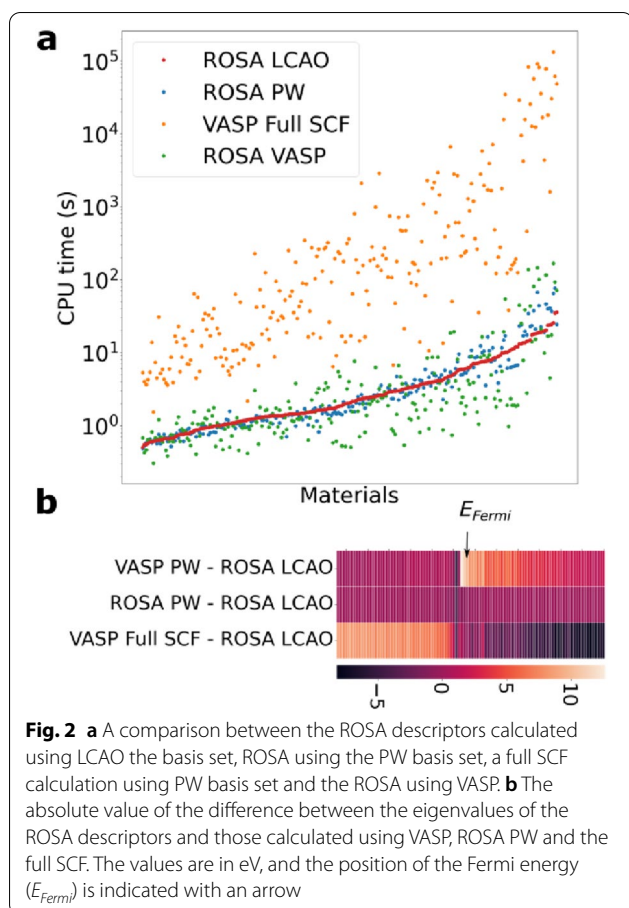


eigenvalues corresponding to 50 occupied and 50 unoccupied orbitals.

The present implementation of the ROSA feature relies on pseudopotentials currently available in the GPAW library, which does not include lanthanides and actinides. If the pseudopotential of any atom in the unit cell is not available, the algorithm substitutes it with that of the yttrium atom. The choice of the yttrium atom is arbitrary, but it had a limited impact on the accuracy of the predictions of target properties in materials where a lanthanide/actinide was substituted with Y.

Basic atom and crystal descriptors (BACD)

This set of descriptors includes the properties of the individual atoms in the crystal, as well as the symmetry information of the crystal structure. The atom-based descriptors include 21 elemental descriptors such as the bulk modulus, ionic radius, rigidity modulus and the molar volume. For each of these descriptors, we include four statistical values over all atoms within the crystal: mean, standard deviation, maximum and minimum values. The crystal structure descriptors include the symmetry group, which is hot-encoded into 230



columns to denote each of the 230 symmetry groups, and the statistical summary of the distance matrix. The distance matrix is the set of distances between each atom and its neighbouring atoms within the crystal. In the present implementation there are 325 BACD features.

Symmetry functions (G)

We calculate the following two symmetry functions for the crystal:

$$G_1^i = \sum_j e^{-\rho(R_{ij}-R_s)^2} f_c(R_{ij})$$

$$G_2^i = \sum_j e^{-\rho(R_{ij}-R_s)^2} e^{-\gamma|Z_i-Z_j|} f_c(R_{ij})$$

$$f_c(x) = \begin{cases} 1, & x < R_c \\ 0, & x \geq R_c \end{cases}$$

where R_{ij} is the distance between the atoms at positions R_i and R_j . The G_1^i and G_2^i symmetry functions are

calculated for each atom i in the crystal, G_1^i is inspired by the function with the same label in Ref. [10], whereas G_2^i is a modification of G_1^i in which a term containing the difference in atomic numbers, $-\gamma|Z_i - Z_j|$, is added to the exponential function. The cutoff function $f_c(R_{ij})$ takes a simpler form than that in Ref. [10]. The variables ρ , R_s , γ and R_c are descriptor parameters. For each crystal structure, the mean value of G_1^i for all atoms i in the crystal represents one descriptor in the feature set. This value is obtained for a specific choice of the variables ρ , R_s , and R_c . Thus, the full descriptor set for the G_1^i function are obtained by assigning different values of ρ , R_s , and R_c and calculating $\sum G_1^i/N$, where N is the number of atoms in the crystal. The same procedure is applied for the G_2^i function, which has the additional parameter γ . In total we generate 600 G descriptors for the subsequent machine learning procedures. Thus the total number of descriptors, including all three classes, is 1,034.

Each of the three descriptor classes is further divided into groups to simplify the analysis. The ROSA descriptors are divided into 3 groups: VBM (the occupied energy levels), CBM (the unoccupied energy levels), and e (the total and constituent energy components including: the total exchange–correlation energy, total kinetic energy, total Hartree energy, Fermi energy, total Coulomb energy, total entropy, total electron-atom interaction energy and total free energy). The BACD descriptors are divided into 5 groups: SG (symmetry group), T (thermal properties of the elements), Geo (geometric features including the distance matrix and the lattice angles) and A (other atomic features). The G descriptors are divided into 2 groups: G1 (the G_1 features) and G2 (the G_2 features). The analysis of the feature importance will involve the aggregation of the individual descriptor groups, as is shown in Fig. 3a.

Results and discussion

Energetics, mechanical and vibrational properties of bulk systems

The ROSA feature includes approximate information about the PBE bandgap and total energy of the system, and therefore, supported by the BACD and G features, an ML model should be able to accurately map them to the converged PBE and formation energy of the system. We construct a dataset of 65,899 materials from MaterialsProject, of which 24,311 are semiconductors (37%). We consider semiconductors as materials with a bandgap >0.1 eV, and metals are otherwise. We train XGBOOST regression and classification models, and for all models we use 80% of the dataset as a training set, and reserve the remaining 20% as a test set.

First we apply the three descriptor classes for the prediction of three energetic properties: the PBE bandgap, formation energy per atom, E_f and the bulk modulus,

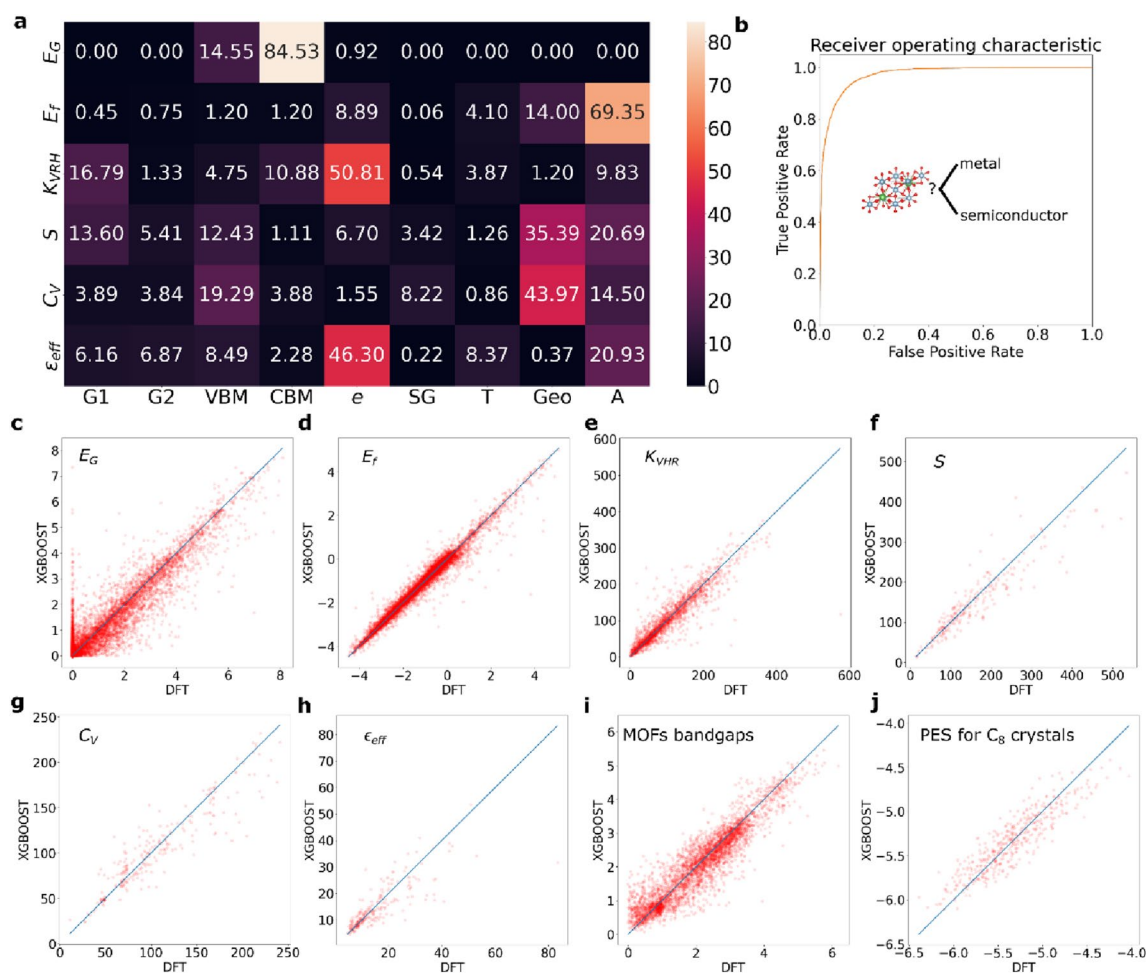


Fig. 3 Performance of the three feature classes (ROSA, BACD and G) in the prediction of the bandgap, E_G , formation energy by atom, E_f , bulk modulus, K_{VHR} , the vibrational entropy, S , the specific heat, C_V and the effective dielectric constant, ϵ_{eff} . **a** The feature importance matrix for predicting the properties by the feature groups outlined in Section “Features”. **b** The receiver operating characteristic (ROC) for the metal/insulator classification model (described in the text). **c–j** The correlation plots for the prediction of the regression models for E_G , E_f , K_{VHR} , S , C_V , ϵ_{eff} , the bandgaps for the materials in the QOMF database and the potential energy surfaces (PESs) for amorphized diamond unit cells, respectively

K_{VHR} . For the PBE bandgap prediction, we train a classifier model to distinguish metallic from semiconducting crystals, and a regression model to predict the bandgap values for the materials. The classifier achieves an area under the curve (AUC) of 0.97. The regression model achieves an $R^2=0.89$ with a mean absolute error (MAE) of 0.22 eV (Table 1). We train another regression model for the prediction of the formation energy per atom, and it achieves a very high accuracy: $R^2=0.97$ with a mean absolute error (MAE) of 0.11 eV. These two accuracy values compare well against those reported in Ref. [8]. The receiver operating characteristic (ROC) of the classifier model is displayed in Fig. 3b, and the correlation plots for the regression models are displayed in Fig. 3c, d.

Table 1 The R^2 and mean absolute error (MAE) values for the prediction of the quantities E_G , E_f , K_{VHR} , S , C_V , ϵ_{eff} , MOF bandgap for the QMOF structures and the total energy for amorphized C_8 crystals (potential energy surface, PES).

Property	Dataset size	R^2	MAE
E_G	65,899	0.89	0.22
E_f	65,899	0.97	0.11
K_{VHR}	13,147	0.86	16
S	1,521	0.85	23
C_V	1,521	0.85	13
ϵ_{eff}	1,521	0.66	3.2
MOF bandgap	20,425	0.86	0.31
PES for C_8 crystals	5,000	0.81	0.14

The MAE values for E_G , E_f and MOF bandgaps are in eV, for K_{VHR} it is in GPa, for S and C_V in meV/atom/K and for the PES for C_8 crystals is eV/atom

The K_{VHR} dataset currently includes 13,147 values in MaterialsProject, and the dataset for the vibrational properties includes 1521 materials in MaterialsProject (which goes down to 1245 materials after removing structures with imaginary phonon frequencies). Upon training an XGBOOST regression model on predicting K_{VHR} values, the model achieves an $R^2=0.86$, MAE=16 GPa (Table 1). While the accuracy of this model is not as high as the model trained on the PLMF features in Ref. [8], which achieved $R^2=0.97$, the accuracy reported here is still a significant one because it covered a larger set of data than the dataset in Ref. [8].

In the feature importance matrix displayed in Fig. 3a, the feature importance is calculated from the *total gain* in the XGBOOST model tree. The key observation in the matrix is that the ROSA descriptors are the most significant contributor to the model prediction of the PBE bandgap E_G , and the third most significant contributor for E_f . As would be expected, the atomic features (A) are the prime contributor for E_f followed by geometric features of the crystal. The primary ROSA descriptor group that contributes to the prediction of E_f is the set of total energy values in the pristine system, e , where as for the prediction of E_G are the sets of VBM and CBM descriptors.

In order to examine the generality of the ROSA descriptors to non-energetic properties, we train regression models to predict three quantities that are obtained from the vibrational spectrum of a material: entropy, S ; specific heat, C_V ; and the effective polycrystalline dielectric function, ϵ_{eff} defined as.

$$\epsilon_{\text{eff}} = \frac{3\epsilon_x\epsilon_y\epsilon_z}{\epsilon_x\epsilon_y + \epsilon_x\epsilon_z + \epsilon_y\epsilon_z},$$

Where ϵ_i is the i component of the dielectric tensor, $i=x,y,z$. The achieved accuracy of the trained regression models are $R^2=0.85$, MAE=23 meV/atom/K; $R^2=0.85$, MAE=13 meV/atom/K and $R^2=0.66$, MAE=3.2, respectively. These values are comparable to those reported in Ref. [26].

The ROSA descriptors group e plays the key role in determining K_{VHR} values, as shown in Fig. 2a; it constitutes the most significant set of descriptors (sum of feature importance of features in the group is ~51%). Owing to the mechanical nature of the bulk modulus, the G1 feature group is the second most significant determinant of the property. The ROSA eigenvalue feature groups VBM and CBM together form the fourth most significant group. These results show the importance of ROSA descriptors for properties that are not directly related to the material's energy eigenvalues or energy components. Those descriptors are even significant in predicting the

three vibrational properties S , C_V and ϵ_{eff} . Given that these properties are strongly related to the material's structure and composition, the Geo, A, G1 and G2 feature groups are the two most significant determinants of all three properties as shown in Fig. 3a. The ROSA eigenvalue group are less significant in determining the three vibrational properties.

We also examine the ability of the ROSA descriptors in capturing the bandgaps in a different class of systems: metal-organic frameworks (MOFs) and amorphized crystals. We trained an XGBOOST model on the entire QMOF database [27], which has 20,425 MOFs along with their PBE-calculated bandgaps. Here we used the ROSA and the BACD descriptors, and achieved a reasonable prediction accuracy, with $R^2=0.86$. This value for R^2 is close to that reported by Rosen et al. [27], where the highest achieved accuracy was $R^2=0.87$ which was obtained using the crystal graph convolutional neural network. For amorphized systems, we trained an XGBOOST model to predict the total energy of an amorphized diamond unit cells (with 8 atoms). The amorphization was performed by running a molecular dynamics in GPAW using the effective medium potential. A dataset of 5,000 snapshots of the carbon system were captured, and a single-point calculation for each of these structures was performed using VASP at the PBE level of theory. The trained model could predict the total energy/atom with an accuracy of $R^2=0.81$. These results assert the applicability of ROSA descriptors across different systems, quantities and calculation methods.

High-level energetics of two-dimensional materials

The largest open-source dataset that provides these bandgaps is C2DB [28], which hosts >4000 2D materials and provides the HSE results for 1,302, GW bandgaps for 357 materials and exciton binding energies for 373 materials. Using this dataset as a training set, Liang and Zhu have recently trained ML models on ~150 2D materials to predict the HSE, GW and BSE energies of ~30 materials in the NREL dataset (which is composed of 3D materials) using a set of descriptors inspired by Phillips's ionicity theory [29], claiming that they demonstrated transferable learning of these bandgap values. The prediction accuracy they reported was quite high when the PBE bandgap is added as a feature.

However, a line of best fit can be obtained for both GW and HSE bandgaps based on PBE bandgaps with R^2 of 0.95 and 0.88, respectively (see Fig. 3a, b). The PBE bandgap therefore plays the role of the ROSA descriptor; it is a bandgap computed with a far less level of theory than the GW and HSE bandgaps. However, with such a strong linear dependence, the PBE bandgap value alongside other features will be the most dominant feature, as can be seen

Table 2 The R^2 and mean absolute error (MAE) values for the prediction of the PBE bandgap (E_{PBE}), the HSE bandgap with (E_{HSE}^{PBE}) and without (E_{HSE}) using the PBE bandgap as a feature, the GW bandgap with (E_{GW}^{PBE}) and without (E_{GW}) using the PBE bandgap as a feature, and the exciton binding energy E_b .

Property	Dataset size	R^2	MAE
E_{PBE}	1302	0.90	0.16
E_{HSE}^{PBE}	1302	0.95	0.23
E_{HSE}	1302	0.86	0.37
E_{GW}^{PBE}	357	0.98	0.26
E_{GW}	357	0.93	0.37
E_b	373	0.89	0.18

The MAE values are in eV

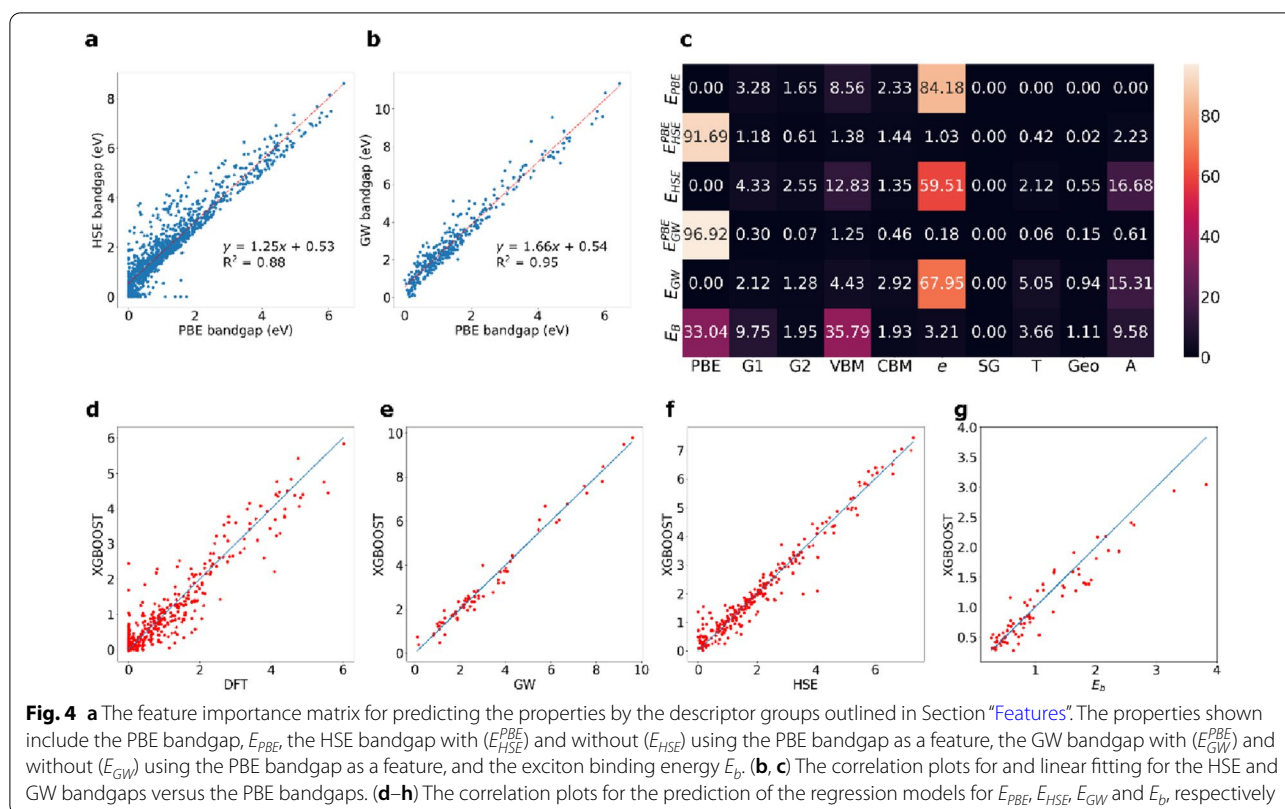
in the SHapley Additive exPlanations (SHAP) plots in Ref. [29].

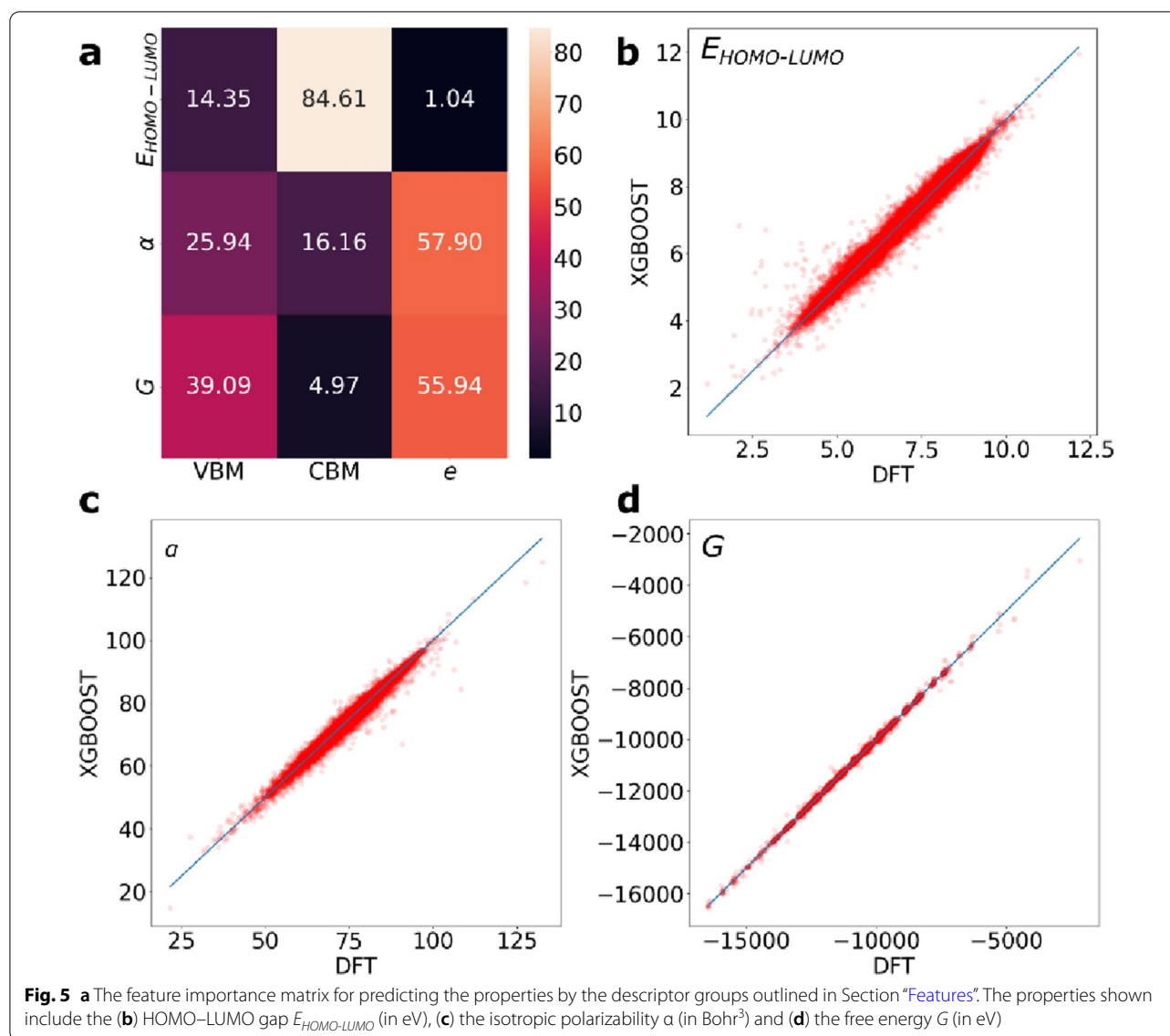
We utilize the three descriptor classes to train regression XGBOOST models to predict the three quantities: HSE and GW bandgaps and the exciton binding energies, E_b . Note that the ROSA features that are used for predicting HSE and GW bandgaps and E_b are the same as those that were used in Section “Energetics, mechanical and vibrational properties of bulk systems”, except that the PBE bandgap is added to the former.

The results are displayed in Fig. 4. For the HSE and GW bandgaps, the prediction accuracy is enhanced with respect to the linear fits for the HSE and GW bandgaps: the regression models achieve $R^2=0.95$, MAE=0.24 eV and $R^2=0.98$, MAE=0.24 eV, respectively (Table 2). The significance of the descriptor groups is displayed in Fig. 3c. As expected, the DFT bandgap is a very significant feature in determining the HSE and GW bandgaps, where the feature importance > 90% (Fig. 4).

When the PBE bandgap is removed as a descriptor from the training sets, the ROSA descriptors become the most significant features for predicting the HSE and GW bandgaps. As can be seen in Fig. 3c, the e descriptor group has the highest significance, followed by the A group. These regression models achieve $R^2=0.86$, MAE=0.37 eV and $R^2=0.93$, MAE=0.37 eV, respectively. The e descriptor group also has the highest significance when we train a model to predict the PBE bandgap values, followed by the A group. The VBM group comes at the third place in terms of significance in predicting the E_{PBE} , E_{HSE} bandgaps and the fourth most significant in predicting the E_{GW} bandgaps.

The E_b does not directly correlate with the PBE bandgap, and therefore the bandgap is not expected to dominate the rest of the descriptors during ML training. The achieved accuracy for predicting E_b is $R^2=0.90$ and





MAE = 0.18 eV (Fig. 3g). According to the aggregated feature importance heatmap in Fig. 3a, the PBE bandgap is the most significant features, but with a significance that is ~34%, unlike the >90% significance in the case of the HSE and GW bandgaps. The VBM replaces e as the most significant set of ROSA descriptors, and the G1 and G2 descriptor groups become more pronounced than in the

Table 3 The R^2 and mean absolute error (MAE) values for the prediction of the HOMO–LUMO gap $E_{HOMO-LUMO}$ (in eV), the isotropic polarizability α (in Bohr³) and the free energy G (in eV)

Property	Dataset size	R^2	MAE
$E_{HOMO-LUMO}$	133,706	0.97	0.15
α	133,706	0.98	0.85
G	133,706	1.00	11

prediction of any of the HSE/GW bandgaps. This shows the importance of adding ROSA descriptors of varying degrees of accuracy to improve the prediction accuracy of a complex energy quantity, E_b . By adding the bandgap of the optimized DFT calculation at the PBE level, as well as the non-self-consistent eigenvalues and energies (ROSA), the aforementioned accuracy was achievable.

Prediction of properties of molecular systems

As the forgoing analysis has been limited to crystal materials, we examine the applicability of the ROSA descriptors to molecular systems. We trained a XGBOOST model on the entire 134 k stable small organic molecules, [30] for the prediction of three molecular quantities: the HOMO–LUMO gap $E_{HOMO-LUMO}$ (in eV), the isotropic

polarizability α (in Bohr³) and the free energy G (in eV). The calculations of these properties were performed at the hybrid exchange DFT/B3LYP level of theory. We build the dataset using only the ROSA descriptors VBM, CBM and e . The XGBOOST model that was trained on 80% of the data achieved a very high prediction accuracy for the three descriptors, as displayed in Table 3: the prediction of $E_{HOMO-LUMO}$ achieved $R^2=0.97$, MAE=0.15 eV; α achieved $R^2=0.97$, MAE=0.15 Bohr³; G achieved $R^2=0.97$, MAE=0.15 eV. The feature importance and correlation plots are displayed in Fig. 5. The CBM descriptors are the key determinants of $E_{HOMO-LUMO}$ as is the case of E_G in Fig. 3a, while e descriptors are the key determinants G , as is the case of E_f in Fig. 3a, as well as α . Thus, ROSA descriptors alone are superior predictors of the electronic properties of molecular systems.

Conclusion

We introduced a novel approach to generate naturally meaningful and computationally efficient features for crystal structures with a small number of descriptors and high predictive power. The robust one-shot ab initio (ROSA) features are calculated by performing only one step of the self-consistent field (SCF) calculation within a density functional theory iteration, and obtaining the eigenvalues and total energies as descriptor values. We demonstrated that ROSA descriptors can accurately predict key material quantities such as the energy bandgap and the bulk modulus. They were particularly a significant factor in accurately predicting the energy bandgap of metal–organic frameworks, the HOMO–LUMO gap of organic molecules and the potential energy surface of amorphized carbon crystals. An ML model trained on ROSA descriptors to predict a bandgap is mainly playing the role of the original, yet computationally expensive, SCF algorithm, rather than a deep learning black box. We also showed that the ROSA features, combined with atom-based and geometry features, are useful for the prediction of thermal properties.

Abbreviations

DFT: Density functional theory; ML: Machine learning; MENA: Meaningful, efficient, small number of descriptors, accurate; PLMF: Property-labelled materials fragments; HOMO: Highest occupied molecular orbitals; LUMO: Lowest unoccupied molecular orbitals; SCF: Self-consistent field; ROSA: Robust one-shot ab initio; HSE: Heyd–scuseria–ernzerhof; GW: An approximation method for the self-energy of many-body systems, where G stands for Green's function, W Coulomb interaction; BSE: Bethe–salpeter equation; LCAO: Linear combination of atomic orbitals basis set; VBM: Valence band maximum; CBM: Conduction band minimum; PW: Plane wave basis set; VASP: Vienna Ab initio simulation package.

Acknowledgements

S. A. T. recognizes the support of the Alfred Deakin Postdoctoral Research Fellowship from Deakin University. This work was supported by the Australian Government through the Australian Research Council (ARC) under the Centre

of Excellence scheme (project number CE170100026). This work was also supported by computational resources provided by the Australian Government through the National Computational Infrastructure National Facility and the Pawsey Supercomputer Centre. This research used resources of the National Energy Research Scientific Computing Center (NERSC), a U.S. Department of Energy Office of Science User Facility located at Lawrence Berkeley National Laboratory, operated under Contract No. DE-AC02-05CH11231.

Author contributions

SAT performed the calculations and data analysis. Both authors discussed the results and wrote the manuscript. Both authors read and approved the final manuscript.

Funding

There is no funding for this work.

Data availability

The XGBOOST machine learning models as well as the scaled test sets are available in <https://github.com/sheriftawfikabbas/crystalfeatures/tree/master/data>.

Code availability

The python code that was used for generating BACD, ROSA and geometry descriptors is available in <https://github.com/sheriftawfikabbas/crystalfeatures>.

Declarations

Ethics approval and consent to participate

Not applicable.

Consent for publication

The authors consent to the publication of this work.

Competing interests

The Authors declare no Competing Financial or Non-Financial Interests.

Author details

¹ARC Centre of Excellence in Exciton Science, School of Science, RMIT University, Melbourne, VIC 3001, Australia. ²Institute for Frontier Materials, Deakin University, Geelong, VIC 3216, Australia. ³Chemical and Quantum Physics, School of Science, RMIT University, Melbourne, VIC 3001, Australia.

Received: 11 April 2022 Accepted: 1 October 2022

Published online: 08 November 2022

References

- Schmidt J, Marques MRG, Botti S, Marques MAL (2019) Recent advances and applications of machine learning in solid-state materials science. *NPJ Comput Mater* 5(1):83. <https://doi.org/10.1038/s41524-019-0221-0>
- Butler KT, Davies DW, Cartwright H, Isayev O, Walsh A (2018) Machine learning for molecular and materials science. *Nature* 559(7715):547–555. <https://doi.org/10.1038/s41586-018-0337-2>
- Jain A, Ong SP, Hautier G, Chen W, Richards WD, Dacek S, Cholia S, Gunter D, Skinner D, Ceder G, Persson KA (2013) Commentary: the materials project: a materials genome approach to accelerating materials innovation. *APL Mater* 1(1):011002. <https://doi.org/10.1063/1.4812323>
- Curtarolo S, Setyawan W, Wang S, Xue J, Yang K, Taylor RH, Nelson LJ, Hart GLW, Sanvito S, Buongiorno-Nardelli M, Mingo N, Levy O (2012) AFLOWLIB.ORG: a distributed materials properties repository from high-throughput ab initio calculations. *Comput Mater Sci* 58:227–235. <https://doi.org/10.1016/j.commatsci.2012.02.002>
- Rudin C (2019) Stop explaining black box machine learning models for high stakes decisions and use interpretable models instead. *Nat Mach Intell* 1(5):206–215. <https://doi.org/10.1038/s42256-019-0048-x>
- Liang J, Zhu X (2019) Phillips-inspired machine learning for band gap and exciton binding energy prediction. *J Phys Chem Lett* 10(18):5640–5646. <https://doi.org/10.1021/acs.jpcllett.9b02232>

7. Legrain F, Carrete J, van Roekeghem A, Curtarolo S, Mingo N (2017) How chemical composition alone can predict vibrational free energies and entropies of solids. *Chem Mater* 29(15):6220–6227. <https://doi.org/10.1021/acs.chemmater.7b00789>
8. Isayev O, Oses C, Toher C, Gossett E, Curtarolo S, Tropsha A (2017) Universal fragment descriptors for predicting properties of inorganic crystals. *Nat Commun* 8:1–12. <https://doi.org/10.1038/ncomms15679>
9. Xie T, Grossman JC (2018) Crystal graph convolutional neural networks for an accurate and interpretable prediction of material properties. *Phys Rev Lett* 120(14):145301. <https://doi.org/10.1103/PhysRevLett.120.145301>
10. Behler J, Parrinello M (2007) Generalized neural-network representation of high-dimensional potential-energy surfaces. *Phys Rev Lett* 98(14):1–4. <https://doi.org/10.1103/PhysRevLett.98.146401>
11. Ward L, Agrawal A, Choudhary A, Wolverton C (2016) A general-purpose machine learning framework for predicting properties of inorganic materials. *NPJ Comput Mater*. <https://doi.org/10.1038/npjcompomats.2016.28>
12. Welborn M, Cheng L, Miller TF (2018) Transferability in machine learning for electronic structure via the molecular orbital basis. *J Chem Theory Comput* 14(9):4772–4779. <https://doi.org/10.1021/acs.jctc.8b00636>
13. Bartók AP, Kondor R, Csányi G (2013) On representing chemical environments. *Phys Rev B* 87(18):184115. <https://doi.org/10.1103/PhysRevB.87.184115>
14. Moriwaki H, Tian YS, Kawashita N, Takagi T (2018) Mordred: a molecular descriptor calculator. *J Cheminform*. <https://doi.org/10.1186/s13321-018-0258-y>
15. Christensen AS, Sirumalla SK, Qiao Z, O'Connor MB, Smith DGA, Ding F, Bygrave PJ, Anandkumar A, Welborn M, Manby FR, Miller TF (2021) Orbnet denali: a machine learning potential for biological and organic chemistry with semi-empirical cost and DFT accuracy. *J Chem Phys*. <https://doi.org/10.1063/50061990>
16. Ramakrishnan R, Dral PO, Rupp M, von Lilienfeld OA (2015) Big data meets quantum chemistry approximations: the Δ machine learning approach. *J Chem Theory Comput* 11(5):2087–2096. <https://doi.org/10.1021/acs.jctc.5b00099>
17. Nandi A, Qu C, Houston PL, Conte R, Bowman JM (2021) Machine learning for potential energy surfaces: a PIP approach to bring a DFT-based PES to CCSD(T) level of theory. *J Chem Phys*. <https://doi.org/10.1063/5.0038301>
18. Hu L, Wang X, Wong L, Chen G (2003) Combined first-principles calculation and neural-network correction approach for heat of formation. *J Chem Phys* 119(22):11501–11507. <https://doi.org/10.1063/1.1630951>
19. Engel E, Dreizler RM (2011) *Density Functional Theory: An Advanced Course*; Springer-Verlag: Berlin. <https://doi.org/10.1007/978-3-642-14090-7>
20. Perdew JP, Burke K, Ernzerhof M (1996) Generalized gradient approximation made simple. *Phys Rev Lett* 77(18):3865–3868. <https://doi.org/10.1103/PhysRevLett.77.3865>
21. Krukau AV, Vydrov OA, Izmaylov AF, Scuseria GE (2006) Influence of the exchange screening parameter on the performance of screened hybrid functionals. *J Chem Phys* 125(22):224106. <https://doi.org/10.1063/12404663>
22. Hybertsen MS, Louie SG (1986) Electron correlation in semiconductors and insulators: band gaps and quasiparticle energies. *Phys Rev B* 34(8):5390–5413. <https://doi.org/10.1103/PhysRevB.34.5390>
23. Albrecht S, Reining L, del Sole R, Onida G (1998) Ab initio calculation of excitonic effects in the optical spectra of semiconductors. *Phys Rev Lett* 80(20):4510–4513. <https://doi.org/10.1103/PhysRevLett.80.4510>
24. Kresse G, Furthmüller J (1996) Efficiency of ab-initio total energy calculations for metals and semiconductors using a plane-wave basis set. *Comput Mater Sci* 6(1):15–50. [https://doi.org/10.1016/0927-0256\(96\)00008-0](https://doi.org/10.1016/0927-0256(96)00008-0)
25. Enkovaara J, Rostgaard C, Mortensen JJ, Chen J, Dulak M, Ferrighi L, Gavnholt J, Glinsvad C, Haikola V, Hansen HA, Kristoffersen HH, Kuisma M, Larsen AH, Lehtovaara L, Ljungberg M, Lopez-Acevedo O, Moses PG, Ojanen J, Olsen T, Petzold V, Romero NA, Stausholm-Møller J, Strange M, Tritsarolis GA, Vanin M, Walter M, Hammer B, Häkkinen H, Madsen GKH, Nieminen RM, Nørskov JK, Puska M, Rantala TT, Schiøtz J, Thygesen KS, Jacobsen KW (2010) Electronic structure calculations with GPAW: a real-space implementation of the projector augmented-wave method. *J Phys Condens Matter*. <https://doi.org/10.1088/0953-8984/22/25/253202>
26. Tawfik SA, Isayev O, Spencer MJS, Winkler DA (2020) Predicting thermal properties of crystals using machine learning. *Adv Theory Simul*. <https://doi.org/10.1002/adts.201900208>
27. Rosen AS, Iyer SM, Ray D, Yao Z, Aspuru-Guzik A, Gagliardi L, Notestein JM, Snurr RQ (2021) Machine learning the quantum-chemical properties of metal-organic frameworks for accelerated materials discovery. *Matter* 4(5):1578–1597. <https://doi.org/10.1016/j.matt.2021.02.015>
28. Hastrup S, Strange M, Pandey M, Deilmann T, Schmidt PS, Hinsche NF, Gjerding MN, Torelli D, Larsen PM, Riis-Jensen AC, Gath J, Jacobsen KW, Jørgen Mortensen J, Olsen T, Thygesen KS (2018) The computational 2D materials database: high-throughput modeling and discovery of atomically thin crystals. *2DMater* 5(4):042002. <https://doi.org/10.1088/2053-1583/aacfc1>
29. Liang J, Zhu X (2019) Phillips-inspired machine learning for band gap and exciton binding energy prediction. *J Phys Chem Lett* 10(18):5640–5646. <https://doi.org/10.1021/acs.jpclett.9b02232>
30. Ramakrishnan R, Dral PO, Rupp M, von Lilienfeld OA (2014) Quantum chemistry structures and properties of 134 kilo molecules. *Sci Data*. <https://doi.org/10.1038/sdata.2014.22>

Publisher's Note

Springer Nature remains neutral with regard to jurisdictional claims in published maps and institutional affiliations.

Ready to submit your research? Choose BMC and benefit from:

- fast, convenient online submission
- thorough peer review by experienced researchers in your field
- rapid publication on acceptance
- support for research data, including large and complex data types
- gold Open Access which fosters wider collaboration and increased citations
- maximum visibility for your research: over 100M website views per year

At BMC, research is always in progress.

Learn more biomedcentral.com/submissions

